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Phase composition and morphology of nanostructured coatings deposited by laser dispersion of a mixture of polyethylene with iron oxalate

Peculiarities of forming of iron oxide coatings with reinforced carbon nanostructures from gas phase generated by laser dispersion of composite target were explored. Influence of technological modes of heat treatment on morphology and phase composition of nanostructured film layers was determined. It was found that on a substrate highly dispersed layers containing carbon nanostructures are formed. Using Raman spectroscopy it was shown that in oxide matrix carbon structures, which are mainly in the form of planar located nanotubes, appear. It was found that with a mass ratio of polyethylene and iron oxalate equal to 1:1, the distribution of the formed nanostructures in size is unimodal with a maximum near 20 nm. At dispersing of polyethylene and iron oxalate mixture with mass ratio 1:2 in deposited layers nanotubes have the least defectiveness. Patterns of influence on morphology and coatings phase composition of relative component abundance in being dispersed by laser radiation composite target were determined. It was shown that with the growing of iron oxalate concentration in the target coating structural heterogeneity increases, subroughness and average size of separate nanostructures in the deposited condensate grow. The obtained polymer matrix nanocomposite films can be used in sensors.

Keywords: polyethylene, oxalate mixture, coating, nanostructures, morphology, phase composition, Raman spectroscopy.

Introduction

One of the promising ways to increase mechanical, thermophysical properties of thin coatings is to reinforce them with nanosized structures, in particular, carbon nanotubes, fullerenes [1–3]. Reinforcement of coatings with carbon nanostructures (nanocrystalline graphite, carbon nanotubes etc) due to their high mechanical properties will allow increasing strength and plasticity of such coatings with simultaneous preservation of high hardness. At the same time, the implementation of such technological technique, as a rule, involves complex, sequentially conducted operations and hence the development of methods for vacuum synthesis of such heterogeneous structures carried out in a single technological cycle with the possibility of controlling the degree of reinforcement, morphology, and phase composition becomes relevant.

It is known that one of the effective methods of nanotubes synthesis is their formation from carbon gases at high enough temperature in the presence of catalysts, for which compounds, as a rule, oxides of transition metals are used [4]. It should be noted that layer deposition of metal oxide is of special interest and implementation of conditions, under which oxides being deposited on the surface as coatings catalyze processes of carbon nanotubes increase being formed in the volume of oxide have scientific and practical interest.

The study considers formation peculiarities of iron oxide coatings with reinforced carbon nanostructures from gas phase generated by laser dispersion of composite target and determines influence of technological modes of heat treatment on morphology and phase composition of nanostructured film layers.

Samples and research methods

Nanostructured coatings using iron oxide were obtained by laser dispersion of composite target based on polyethylene and iron oxalate, preheated to the temperature 330 °C. The target was prepared by polyethylene powder (PP) and iron oxalate ($\text{FeC}_2\text{O}_4 \cdot 2\text{H}_2\text{O}$) mixing with various mass ratio 1:1, 1:2, 1:3. The received powders mixture was placed into crucible on the heating panel. When the pressure in the vacuum chamber was $\sim 4 \cdot 10^{-3}$ Pa the power of the heating panel turned on providing the target heating up to 330 °C. Laser L-2137U+HG-5 was used as a source of laser radiation, which enables to carry out wavelength agility of laser radiation within the interval from infra-red (1064 nm) to ultraviolet range (355, 266 nm). Frequency of

impulse rate was 10 Hz, the duration of pumping impulse in Q-switched mode was 6 ns. The laser spot diameter made 7 mm. Radiation with wavelength of $\lambda = 532$ nm was used in the study. Pulse energy corresponded to 380 mJ.

The source of laser radiation switched on only after the formation of homogenous melt in the crucible. The impact of laser radiation with melt was accompanied by intensive gas phase generation. The crucible heating wasn't turned off within the whole period of laser dispersion. The substrate temperature was 350 °C that was considerably lower than the temperature of the synthesis of carbon nanostructures from the carbons on the surface of the catalyst which makes 500–700 °C [1–4].

Experimental results and their discussion

Patterns of influence on morphology and coatings phase composition of relative component abundance in being dispersed by laser radiation composite target were determined. It was shown that with the growing of iron oxalate concentration in the target coating structural heterogeneity increases, subroughness and average size of separate nanostructures in the deposited condensate grow (Fig. 1, Table 1).

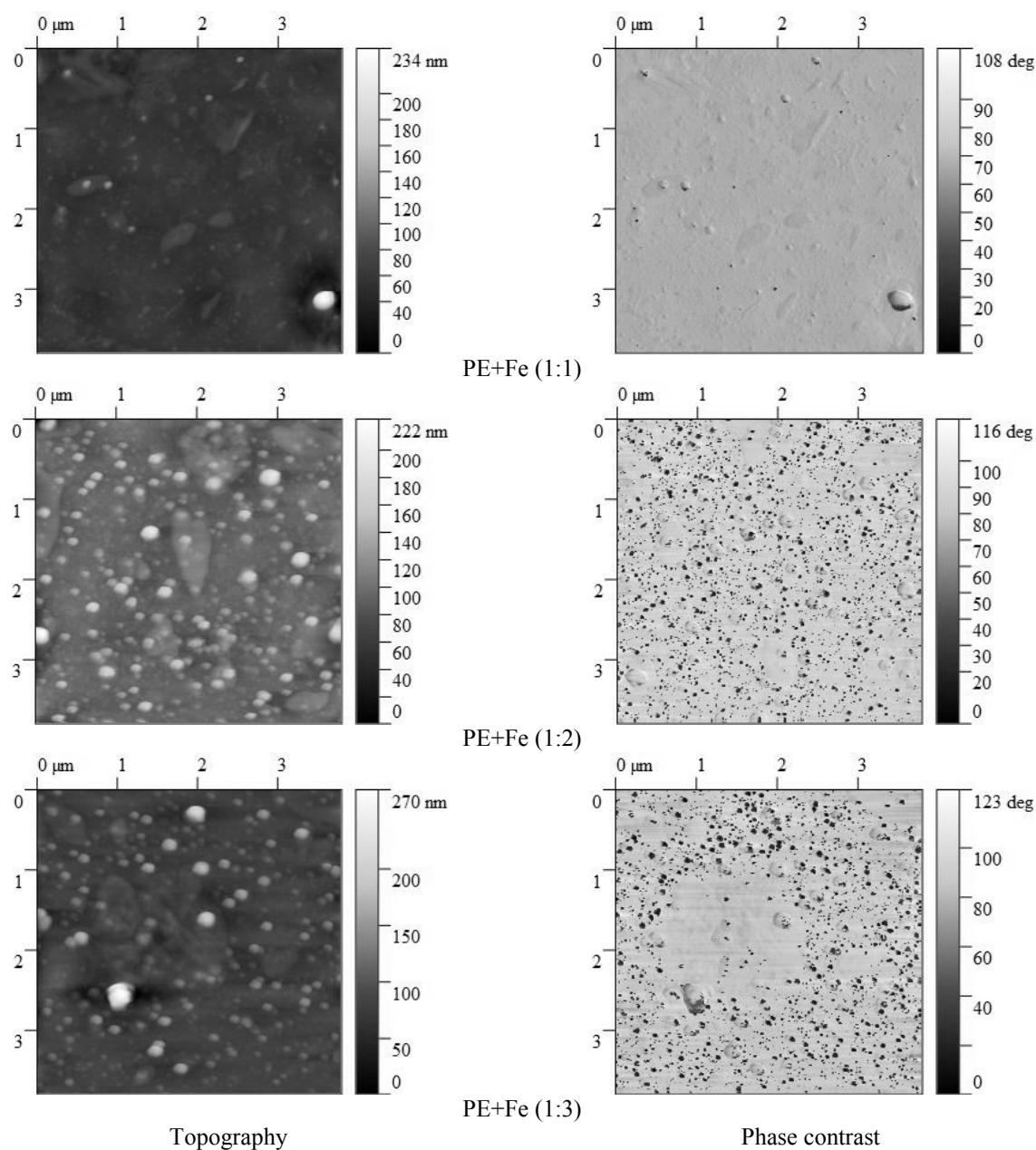


Figure 1. AFM images of deposited metal carbon films of the condensate with mass ratio (1:1, 1:2 and 1:3) of the target components (PE+Fe)

Influence of mass ratio of the target (PE+Fe) on morphological peculiarities of deposited metal-carbon condensate

Coating	Mass ratio	Medium height, nm	R_{ms} , nm	Grain density, pc.	Average diameter of grains, nm
PE+Fe	1:1	59.3/139.4	11.0/22.6	246/94	50/91
	1:2	84.0/64.6	14.9/15.3	255/138	70/112
	1:3	98.5/39.2	18.6/6.5	244/85	77/52

Note. Without thermal annealing/thermal annealing.

Particles on the surface contain intermediate compounds of molecule decomposition of iron oxalate and their size increase can be explained by their more intensive dissociation with phase formation of various compositions. Besides, it must be taken into consideration that laser dispersion is conducted in the preheated to 330 °C mixture of polyethylene and iron oxalate. According to [5] iron oxalate hydrant loses crystallization water in the vacuum at 142 °C. At further heating to 400–500 °C due to laser treatment, decomposition products consist of iron oxide Fe_3O_4 or FeO , containing pure iron. As on the figures of phase contrast (Fig. 1) structural heterogeneity is darker, they can be interpreted as objects with lower hardness consisting of iron mainly.

It was found that with a mass ratio of polyethylene and iron oxalate equal to 1:1 the distribution of the formed nanostructures N in size (radius R_N) is unimodal with a maximum near 20 nm (Fig. 2). With the further increase in the content of iron oxalate (1:2, 1:3) the radius distribution of the formed nanostructures shifts towards large particles and with high iron oxalate content in the target becomes unimodal, local maxima appear in the range from 30 to 50 nm, indicating more intensive emission of nanophase iron.

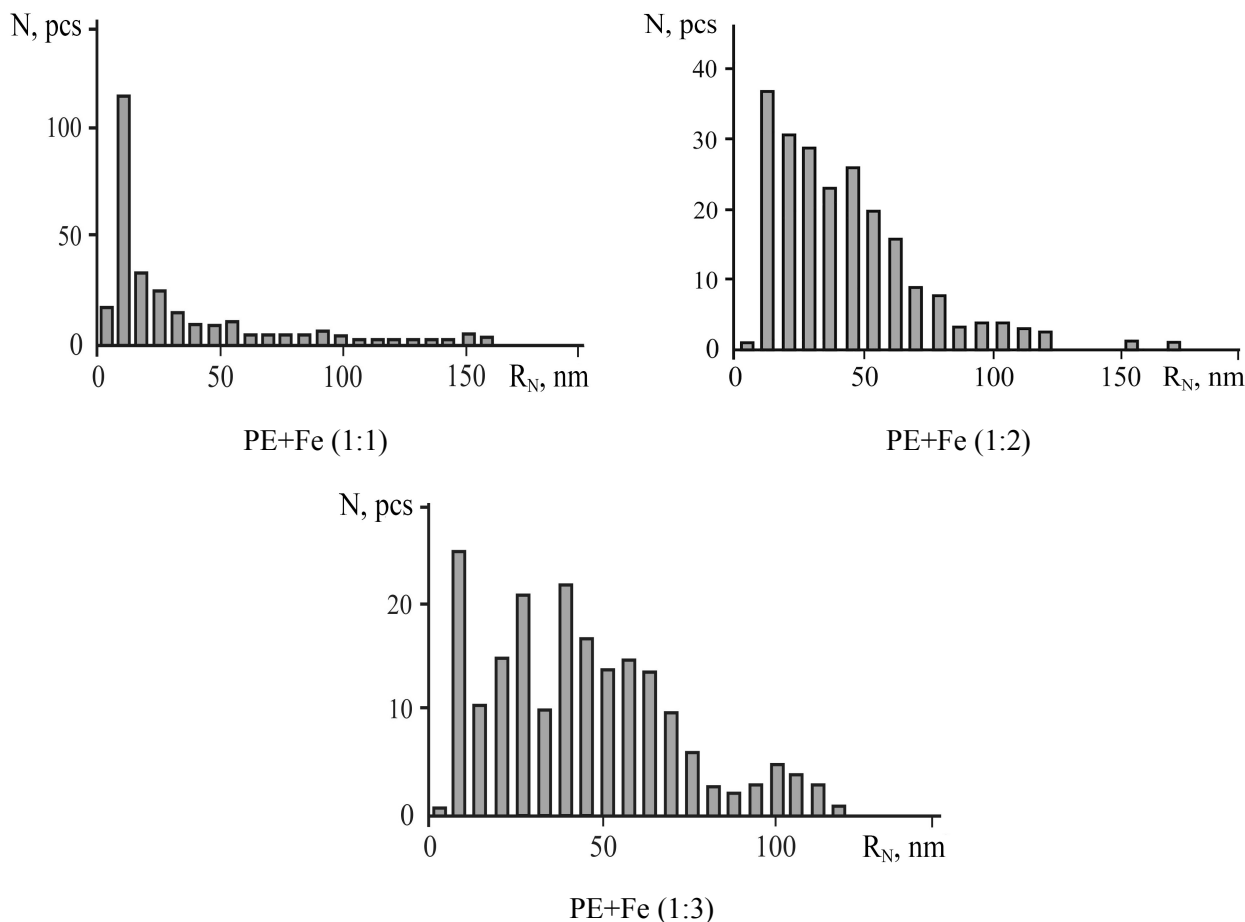


Figure 2. Distribution histogram of the forming nanostructures N in their radius size R_N with mass ratio (1:1, 1:2 and 1:3) of the target components (PE+Fe)

Heat treatment of coatings during an hour in air at a temperature of 500 °C, in case of mass ratio polyethylene-iron oxalate 1:1 and 1:2, leads to the density reduction of structural formations and their size R_N increase (Fig. 3, 4 and Table 1). The process is considerably stemmed from the occurrence of coalescence and partial oxidation of pure iron particles. At the same time in case of dispersion of the target with high content of iron oxalate (1:3) the distribution of the particles in size transforms from unimodal into exponential. Their size decrease is observed which is probably related with the intensive occurrence of iron oxidation. The conclusion is supported by significant (more than 3 times) decrease of particles density (Table 1).

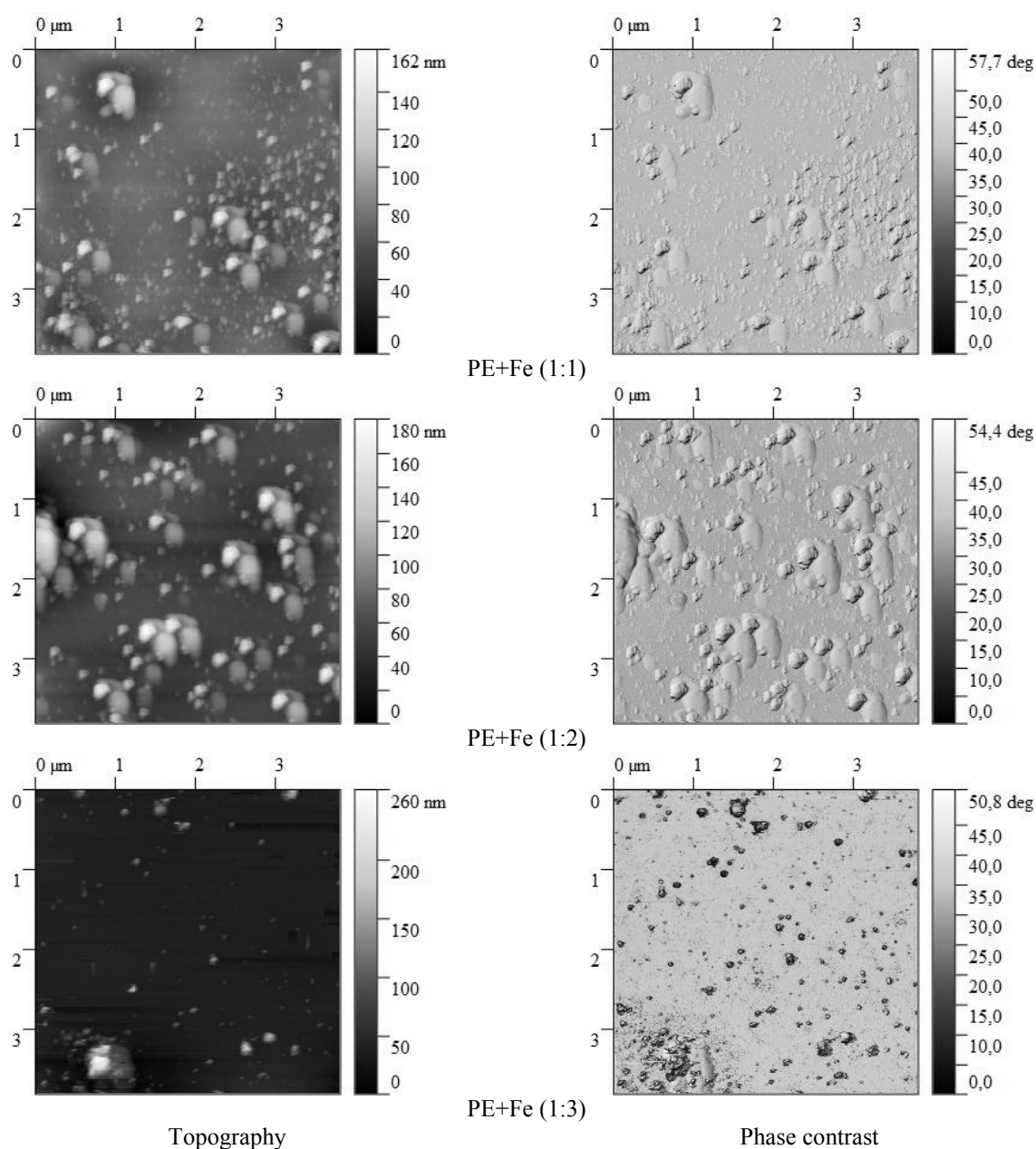


Figure 3. AFM images of deposited metal carbon films of the condensate after the thermal annealing at a temperature of 500 °C with mass ratio (1:1, 1:2 and 1:3) of the target components (PE+Fe)

On Figure 5 Raman spectra of nanostructured layers deposited by laser dispersion of the target with various content are shown. According to [6–10] spectra of such form are typical for carbon nanomaterials. It is known that vibration band (G-peak) at 1575 cm^{-1} is determined by stretching vibration mode E_{2g} of graphite lattice of infinite crystal [11]. The mode at 1355 cm^{-1} (D-peak) Raman spectrum corresponds to A_{1g} mode, radial modes of hexagonal rings (often called «breathing» mode) [12]. D-peak occurs as a result of a binary resonant process [13] and appears if there are defects of structured carbon [14] and evidently marked bound-

aries of nanocrystalline areas, it is typical for various carbon nanostructures (single-walled and multi-walled nanotubes, fullerenes, graphenes etc.) [15].

It should be mentioned that the band, determined by symmetric modes of C–H bonds in CH₂ and asymmetric ones in CH₃ [7], is also registered in the interval between 2800–3000 cm⁻¹. C–H bonds in the coating result from the dispersion of polyethylene and adsorption of volatile products on the substrate surface.

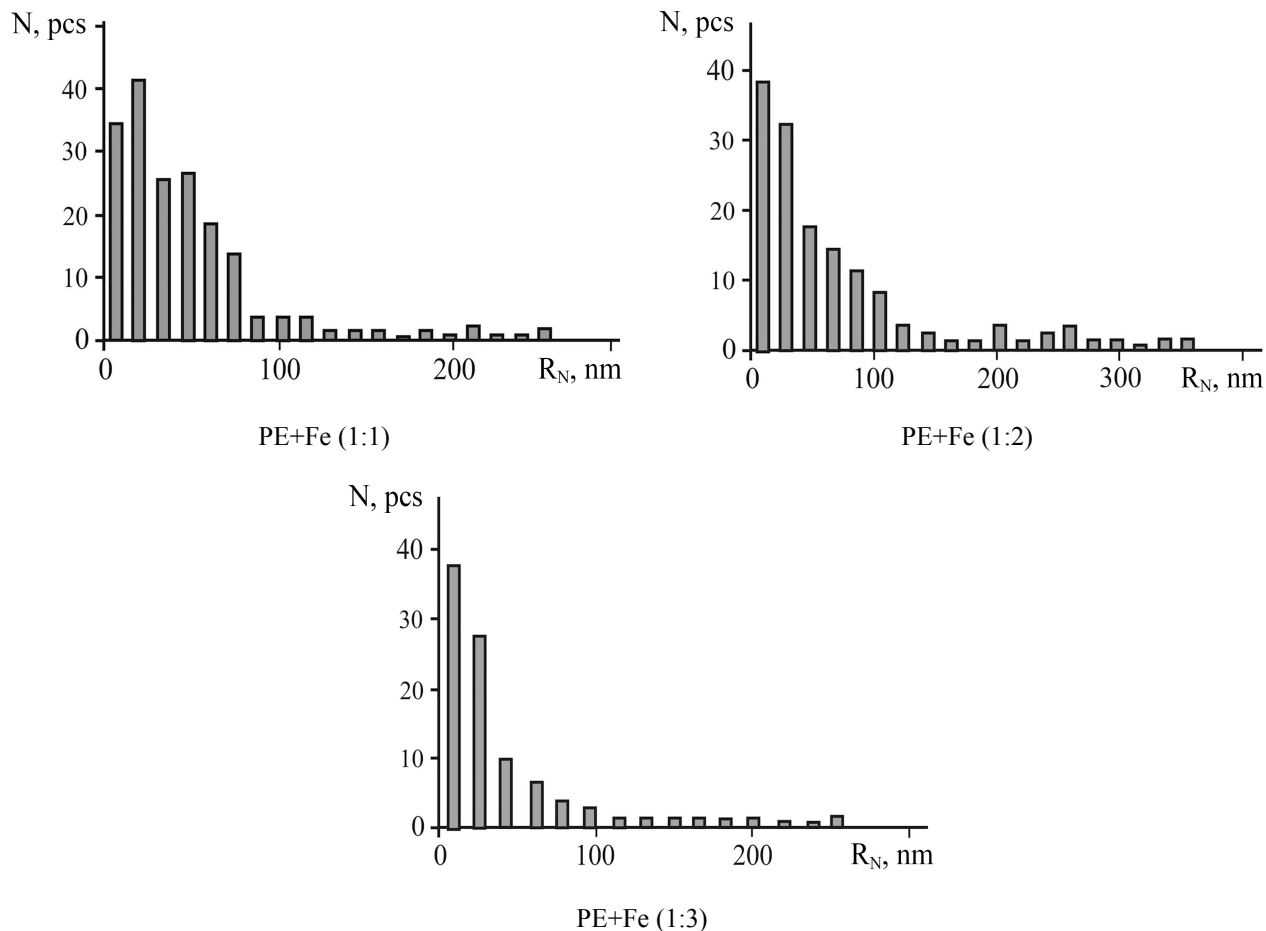


Figure 4. Distribution histogram of the forming nanostructures N in their radius size R_N after the thermal annealing at a temperature of 500 °C with mass ratio (1:1, 1:2 and 1:3) of the target components (PE+Fe)

The presented Raman spectra clearly indicate that resulting from laser dispersion of a mixture of polyethylene with iron oxalate in the volume of the layer, carbon nanostructures such as fullerenes, nanotubes and probably multi-walled nanotubes (MWNT) are formed. Except E_{2g} and A_{1g} the occurrence of radial modes near 200 cm⁻¹ [16] and G'-peak in the range of 2450–2650 cm⁻¹ [17] is characteristic of single-walled nanotubes. Spectra do not allow identifying them clearly as they have only E_{2g} and A_{1g} modes which are typical for carbon coatings and materials. Aiming at analysis of dispersion influence on the content of carbon formations, the obtained Raman spectra were arranged according to Gaussian elimination on D- (~ 1350 cm⁻¹) and G-peaks (~ 1580 cm⁻¹). Besides, base on requirements of error minimization, at decomposition extra D1-peak near 1500 cm⁻¹ (Fig. 5), associated with defective packaging and fullerenes and graphene clusters [8] was added.

The intensity ratio of I_G and I_D , is usually used to determine defect rates of structures [7, 10]. For example, if I_D is higher than I_G , this points to the failure of long-range crystalline and disordered areas. Such effect can be observed not only in nanotubes with a large number of defects, but also can appear due to metal impurities in amorphous carbon particles, short carbon fiber and simply due to layers of amorphous carbon on external surfaces of nanotubes [7].

Among examined testing samples layers deposited by dispersion of polyethylene with iron oxalate mixture with mass ratio 1:2, are characterized by the minimum value of I_D/I_G ratio (Table 2) which combined

with the smallest width of D1-peak indicates minimum content of nanostructures defects. G-peak discomposition to 1600 cm^{-1} also demonstrates disorderliness of crystalline lattice of graphite [9]. Besides, indicated samples are characterized by great D-peak width value showing high content of nanoclusters of sp^3 -hybridized carbon [18].

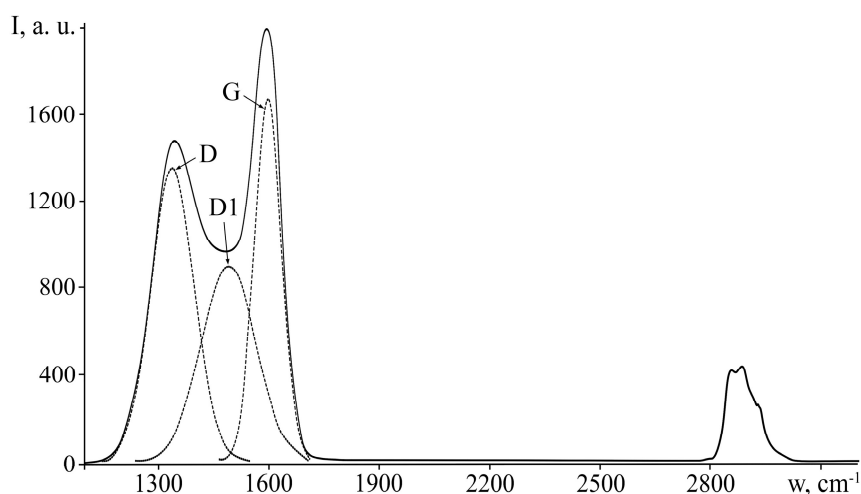


Figure 5. Typical Raman spectrum for nanostructured coatings deposited by laser dispersion of a mixture of polyethylene with iron oxalate

At the same time D1-peak shifting to 1520 cm^{-1} is typical for samples deposited by dispersion of polyethylene with iron oxalate mixture with mass ratio 1:3, which combined with I_D/I_G increase, the decrease of the size of graphene clusters can be explained [8], since under [11] I_D/I_G is in inverse proportion to the size of carbon clusters. Nanostructures (nanotubes, fullerenes or graphene fragment) in condensate deposited by laser dispersion of polyethylene with iron oxalate are indirectly confirmed by Raman spectrum of carbon coatings (Table 2) deposited by laser dispersion of graphite.

Table 2

Phase composition of nanostructured coatings

Mass ratio PE+Fe	D-peak		D1-peak		G-peak		I_D/I_G
	Position, cm^{-1}	Width, cm^{-1}	Position, cm^{-1}	Width, cm^{-1}	Position, cm^{-1}	Width, cm^{-1}	
УП	1430	245	–	–	1562	200	0,46
1:1	1330	138	1479	222	1599	90	0,54
1:2	1348	180	1480	105	1590	97	0,60
1:3	1348	148	1520	149	1602	88	1,01

Firstly, D-peak of carbon coatings is near 1430 cm^{-1} , and D-peak of nanostructured carbon layers is near 1350 cm^{-1} . Such displacement is caused by structure disorder to a greater degree that is evident as carbon coatings are an amorphous carbon matrix [19]. Secondly, Raman spectra decomposition of carbon coatings excludes variants of peak arrangement near 1500 cm^{-1} , related to defective packaging and graphene clusters.

After condensate heat treatment deposited by laser dispersion of polyethylene with iron oxalate due to the occurrence of processes of coalescence on Raman spectrum peaks appear near 220 cm^{-1} , 280 cm^{-1} , 392 cm^{-1} , 601 cm^{-1} , corresponding to $\alpha\text{-Fe}$ and iron carbide [20], and also the peak near 650 cm^{-1} , caused by Fe_3O_4 can be seen [21]. Peaks occurrence near 520 cm^{-1} and 960 cm^{-1} , relating with polycrystalline silicon [22], is explained by lack of coating density. After heat treatment the peak near $2800\text{--}3000\text{ cm}^{-1}$, as well as D- and G- peaks disappear caused by destruction of C–N bonds and carbon oxidation in the air.

Conclusion

It was shown that upon laser dispersion of a composite target based on polyethylene and iron oxalate, preheated to a temperature of $330\text{ }^\circ\text{C}$, coatings consisting of iron and its oxides and reinforced with carbon nanostructures are deposited. It was found that with a mass ratio of polyethylene and iron oxalate equal to 1:1, the distribution of the formed nanostructures in size (radius) is unimodal with a maximum near 20 nm.

With a further increase in the content of iron oxalate (1: 2, 1:3), the radius distribution of the formed nanostructures becomes multimodal, local maxima appear in the range from 30 to 50 nm. During heat treatment of coatings in air at a temperature of 500 °C, the density of structural formations decreases and their size increases which is explained by the occurrence of processes of coalescence and partial oxidation of pure iron particles. Using Raman spectroscopy it was found that carbon nanostructures formed in the coating volume are highly defective and the phase composition depends on the content of polyethylene in the target. The obtained polymer matrix nanocomposite films can be used in sensors [23, 24].

The study was supported by the Ministry of Education of the Republic of Belarus (task 2.64), as well as by the Ministry of Science and Higher Education of Russian Federation (project No. 0706–2020–0022).

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Полиэтиленнің темір оксалатымен қоспасын лазерлік диспергирлеу арқылы тұндырылатын наноқұрылымды жабындардың фазалық құрамы мен морфологиясы

Композиттік нысанның лазерлік дисперсиясы нәтижесінде пайда болатын газ фазасынан күшейтілген көміртекті наноқұрылымдары бар темір оксиді жабындарының қалыптасу ерекшеліктері зерттелген. Термиялық өндеудің технологиялық режимдерінің наноқұрылымды пленка қабаттарының морфологиясы мен фазалық құрамына әсері нақтыланған. Төсеніште көміртегі наноқұрылымдары бар жоғары дисперсті қабаттар пайда болатындығы дәлелденген. Раман спектроскопиясы әдісімен оксид матрицасында көміртекті құрылымдар пайда болатындығы көрсетілді, олар негізінен жалпақ орналасқан нанотүтікшелер түрінде ұсынылған. Полиэтилен мен темір оксалатының 1: 1 массалық қатынасында қалыптасқан наноқұрылымдардың мөлшері бойынша таралуы шамамен 20 нм болатын унимодальды екендігі анықталды. 1: 2 массалық қатынасы бар полиэтилен мен темір оксалатының қоспасын дисперсиялаған кезде, нанотүтікшелер балқытылған қабаттарда ең аз ақауларға ие болады. Лазерлік сәулеленудің композициялық нысанын дисперсиялау кезінде компоненттердің салыстырмалы құрамы жабындардың морфологиясы мен фазалық құрамына әсер ету заңдылықтары анықталған. Мақсатты жабындағы темір оксалатының концентрациясының жоғарылауымен құрылымдық гетерогенділік жоғарылайды, ішкі кедір-бұдыр және босатылған конденсаттағы жеке наноқұрылымдардың орташа мөлшері артады. Алынған полимерлі матрицасы бар нанокомпозитті пленкаларды сенсорларда қолдануға болады.

Кілт сөздер: полиэтилен, темір оксалат, жабын, наноқұрылымдар, морфология, фазалық құрам, раман спектроскопиясы.

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Фазовый состав и морфология наноструктурированных покрытий, осаждаемых лазерным диспергированием смеси полиэтилена с оксалатом железа

Исследованы особенности формирования покрытий оксида железа с армированными углеродными наноструктурами из газовой фазы, генерируемой лазерным диспергированием композитной мишени. Определено влияние технологических режимов термообработки на морфологию и фазовый состав наноструктурированных пленочных слоев. Обнаружено, что на подложке формируются высокодисперсные слои, содержащие углеродные наноструктуры. Методом рамановской спектроскопии было показано, что в оксидной матрице возникают углеродные структуры, которые в основном представлены в виде плоскорасположенных нанотрубок. Установлено, что при массовом соотношении полиэтилена и оксалата железа 1: 1 распределение сформированных наноструктур по размерам является унимодальным с максимумом около 20 нм. При диспергировании смеси полиэтилена и оксалата железа с массовым соотношением 1: 2 в наплавленных слоях нанотрубки имеют наименьшую дефектность. Определены закономерности влияния на морфологию и фазовый состав покрытий относительного содержания компонентов при диспергировании композитной мишени лазерного излучения. Показано, что с увеличением концентрации оксалата железа в целевом покрытии повышаются структурная неоднородность, внутренняя шероховатость и средний размер отдельных наноструктур в выпавшем конденсате. Полученные нанокомпозитные пленки с полимерной матрицей могут быть использованы в сенсорах.

Ключевые слова: полиэтилен, оксалат железа, покрытие, наноструктуры, морфология, фазовый состав, КР-спектроскопия.

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