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B.D. Igamov<sup>1</sup>, G.T. Imanova<sup>2\*</sup>, A.I. Kamardin<sup>1</sup>, I.R. Bekpulatov<sup>3</sup>

<sup>1</sup>Scientific and technical center with a design bureau and pilot production of the Academy of Sciences of the Republic of Uzbekistan,

Tashkent, Uzbekistan;

\*2Ministry of Science and Education of the Republic of Azerbaijan, Institute of Radiation Problems, Baku, Azerbaijan; <sup>3</sup>Tashkent State Technical University, 100095, Uzbekistan (\*E-mail: gunel\_imanova55@mail.ru)

# Formation of targets and investigation of Mn<sub>4</sub>Si<sub>7</sub> coatings produced by magnetron sputtering

The morphology, composition, electrical and optical properties of bulk samples and vacuum coatings of Mn<sub>4</sub>Si<sub>7</sub> obtained by magnetron sputtering on a SiO<sub>2</sub>/Si structure were studied. It is shown that manganese silicide coatings with a thickness of about 150 nm are close in properties to bulk Mn<sub>4</sub>Si<sub>7</sub>, have a uniform finegrained structure of a semiconductor nature, which is characterized by thermal sensitivity up to 20-30  $\mu$ V per degree. In addition, this article presents the electrophysical properties of high manganese silicide films produced by the authors by magnetron sputtering method. Heated films Mn<sub>4</sub>Si<sub>7</sub> -146 nm coating has a uniform structure with fine grains, due to sufficient coating density. Since Mn<sub>4</sub>Si<sub>7</sub> nanoclusters are semiconductor materials, it can be assumed that there will be energy barriers for charge carriers at the nanocluster–amorphous phase interface separating this phase. An increase in thermal sensitivity from 0  $\mu$ V/K to 20  $\mu$ V/K up to 800 K is explained by the disappearance of energy barriers for charge carriers at the nanocluster–amorphous phase interface due to the ordering of nanoclusters. The change from 20  $\mu$ V/K to 28  $\mu$ V/K upon cooling is explained by the appearance of structural relaxation in the amorphous phase.

*Keywords:* Hall constant, Mn<sub>4</sub>Si<sub>7</sub>, thin coating, nanocluster, electrical conductivity, nanostructure, resistivity, volume concentration.

### Introduction

The main task facing scientists all over the world today is to search for environmentally friendly types of energy and increase the utilization rate of identified types. The main goal at the same time is to receive energy without harming the environment. Unfortunately, the efficiency of currently produced thermo- and photo batteries is very low. A key role in solving this problem is played by the creation of new materials and structures or the replacement of existing ones with cheap and high-quality ones.

Receiving and converting energy is one of the most important activities of modern civilization [1-7]. In this regard, much attention is paid to solid-state thermoelectric converters that do not have moving parts, operate silently, have high reliability and small size. An increase in the efficiency of using thermoelectric materials is associated with the formation of high quality layers [8-23]. Of all silicon compounds of thermoelectric interest, one can choose compounds representing a certain class of materials. These are, for example, solid solutions based on cobalt monosilicide (CoSi), high manganese silicide (MnSi<sub>1.7</sub>), and Mn<sub>2</sub>X (X=Si, Ge, Sn). Highmanganese silicide (HMS-MnSi<sub>1.7-1.75</sub>), even in the unalloyed state, has a high thermoelectric efficiency and is a good basis for creating an efficient *p*-type thermoelectric. Therefore, thin vacuum coatings of Mn<sub>4</sub>Si<sub>7</sub> were chosen as objects of study.

### Experimental

To obtain thin-film samples of  $Mn_4Si_7$ , a disk target was first formed. Pure monocrystalline silicon and manganese were first pulverized in a mill (HERZOG HSM-100P), then 52.9 % Mn and 47.1 % Si (by mass) were mixed and sintered using electric spark plasma welding (SPS). The  $Mn_4Si_7$  disk target was pressed in a setup under vacuum conditions with a residual gas pressure of  $10^{-2}$  Torr at a temperature of 1050 °C, with a pressing force of  $6.5 \cdot 10^4$  N for 2 hours (Fig. 1).



Figure 1. Preparation of the Mn<sub>4</sub>Si<sub>7</sub> target by the (SPS) method

Polished silicon wafers of the Si(111) type with a diameter of 60 mm were used as a base (substrate) for deposition of thermally sensitive coatings. A group of plates after preliminary chemical cleaning in an ammonium peroxide solution, washing and drying were subjected to high-temperature treatment to create an oxide layer. Silicon dioxide layers of various thicknesses were grown on plates in an environment of dry oxygen at a temperature of  $1200 \pm 5$  °C in a diffusion furnace of the SDO-125 type. The SiO<sub>2</sub>/Si structures prepared in this way were processed in a vacuum working chamber. The surface of the Si(111) substrate was cleaned with an Ar plasma flow for 1 minute. The device and the process of processing plates is shown in Figure 2.

The SiO<sub>2</sub>/Si structures were placed in a modified (EPOS-PVD-DESK-PRO) installation for magnetron sputtering of the Mn<sub>4</sub>Si<sub>7</sub> target and coating formation. The coating formation process was carried out after reaching the starting vacuum degree of about  $10^{-5}$ Torr. The SiO<sub>2</sub>/Si structures were treated individually with heating up to 150 °C. The pressure of the working gas (pure argon) during spraying was (2-4)·10<sup>-3</sup>Torr. The discharge current was 200–300 mA at voltages of 450–550 V. The coating deposition time was 2–10 minutes. In one vacuum cycle, 3 structures were sequentially processed. The morphology, microstructure, and chemical composition of the coated samples were determined by scanning electron microscopy and energy-dispersive X-ray spectroscopy (Scios FEI; Quanta 200 3D setups). The electrical properties of the Mn<sub>4</sub>Si<sub>7</sub> coating were studied by a four-probe method (JANDEL RM3000 setup), and the Hall constant was determined using an ECOPIA setup (HMS-3000 VER3.53).

 $Mn_4Si_7$  samples obtained by sintering (SPS) and coated samples obtained by magnetron sputtering on the EPOS-PVD-DESK-PRO facility were studied.



Figure 2. The process of forming the Mn<sub>4</sub>Si<sub>7</sub> coating by the method of magnetron sputtering

## Results and Discussion

Figure 3a, b shows images of the surface of the samples and the results of energy dispersive analysis: (a) — sample prepared by the (SPS) method, (b) — sample obtained by magnetron sputtering.

Figure 4 shows an electron microscope photograph of a cut of the  $Mn_4Si_7$  coating on the  $SiO_2/Si(111)$  structure. The measurements show a thickness of the silicon dioxide coating of about 249 nm, and the thickness of the  $Mn_4Si_7$  coating is about 146 nm.

The results of averaged measurements by the four-probe method of the conductivity of  $Mn_4Si_7$  (SPS) samples and the conductivity of  $Mn_4Si_7$  coatings obtained by magnetron sputtering are shown in Table 1.

Table 1

Sample type	Sheet resistance values, Ohm/square	Average value of sheet resistance, Ohm/square
Volume	508-602	556
Coating	4380-4460	4401

Layer resistances of bulk samples Mn4Si7 and Mn4Si7 coatings on the SiO2/Si structure





![](_page_3_Figure_2.jpeg)

Figure 3. Image of Mn<sub>4</sub>Si<sub>7</sub> sample (Scios FEI; Quanta 200 3D)

![](_page_3_Picture_4.jpeg)

Figure 4. Image of nanosized Mn<sub>4</sub>Si<sub>7</sub> coating on SiO<sub>2</sub>/Si structure

The measurement results show that the resistance of the samples (SPS) is different, which is possibly due to the different concentrations of Mn and Si in different zones of the samples. In  $Mn_4Si_7vacuum$  coatings, the uniformity of the formed layer is higher, which indicates the same coating thickness.To determine the Hall constant on an ECOPIA(HMS-3000 VER3.53) instrument, a (SPS)  $Mn_4Si_7$  sample with an area of 1 cm<sup>2</sup> and a thickness of 1 mm and a sample with a vacuum coating of  $Mn_4Si_7$  with a thickness of 146 nm were used. In all measurements, the current strength was 100  $\mu$ A, the magnetic field induction was 0.54 T, and the temperature was 27 °C.

The results of measurements obtained for these samples (Table 2), the dependences of the samples (VAX) and resistance on current (Fig. 5) are presented. There are the results of Hall measurements of bulk samples of  $Mn_4Si_7$  and  $Mn_4Si_7$  in the SiO<sub>2</sub>/Si structure in Table 2.

#### Table 2

Options	Volume Mn <sub>4</sub> Si <sub>7</sub>	Coating Mn <sub>4</sub> Si <sub>7</sub>
Resistivity, Ohm cm	7.826 • 10 -4	6.409.10-4
Hall constant, cm <sup>3</sup> /C	$1.285 \cdot 10^{-3}$	$1.597 \cdot 10^{-3}$
Conductivity, 1/Ohm·cm	$1.278 \cdot 10^3$	$1.560 \cdot 10^3$
Surface concentration, cm <sup>-2</sup>	$1.215 \cdot 10^{17}$	$9.770 \cdot 10^{16}$
Volume concentration, cm <sup>-3</sup>	$4.859 \cdot 10^{21}$	$3.908 \cdot 10^{21}$
Carrier mobility, $cm^2/V \cdot s$	1.642	2.492

#### Electrophysical properties of high manganese silicide structure formed by two different methods

From the results obtained with the ECOPIA instrument (HMS-3000 VER3.53), it can be seen that the bulk sample (SPS) and the coating sample (EPOS-PVD-DESK-PRO) are close to each other.

Figure 6 shows the optical absorption and transmission spectra of the samples obtained on an IR Tracer-100-SHIMADZU instrument.

The measurement results show that the nanoscale  $Mn_4Si_7$  coating (a) has an IR transmission of 35 %, (b) an IR absorption of about 1 %, from which it can be seen that the  $Mn_4Si_7$  coating has a low IR absorption of the rays. Figure 6 shows the results of measurements of the Seebeck coefficient (*S*) and resistance (*R*) of the  $Mn_4Si_7$  -146 nm coating during heating and cooling.

![](_page_4_Figure_7.jpeg)

Figure 5. a) samples (I-V), b) relationship between (I-R)

![](_page_5_Figure_1.jpeg)

Figure6. Spectra obtained with the IRTracer-100 - SHIMADZU instrument

![](_page_5_Figure_3.jpeg)

Figure 7. Temperature dependence of the Seebeck coefficient (S) and resistance (R) of a thin Mn<sub>4</sub>Si<sub>7</sub> coating

When heated, Mn<sub>4</sub>Si<sub>7</sub> -146 nm coating has a uniform structure with fine grains, due to sufficient coating density. With an increase in temperature from room temperature to 800 K, a decrease in resistance is observed from 250 Ohms to 25 Ohms, which in turn indicates a change in the characteristics of this material. The Seebeck coefficient decreases upon heating from room temperature from 300 K to 620 K. Since Mn<sub>4</sub>Si<sub>7</sub> nanoclusters are semiconductor materials, it can be assumed that there will be energy barriers for charge carriers at the nanocluster–amorphous phase interface separating this phase (Fig. 7).

An increase in thermal sensitivity from 0  $\mu$ V/K to 20  $\mu$ V/K up to 800 K is explained by the disappearance of energy barriers for charge carriers at the nanocluster–amorphous phase interface due to the ordering of nanoclusters. The change from 20  $\mu$ V/K to 28  $\mu$ V/K upon cooling is explained by the appearance of structural relaxation in the amorphous phase. Seebeck coefficient (*S*) and resistance (*R*) vary with coating thickness.

#### Conclusions

Studies of the parameters of thin coatings of manganese silicide deposited on the SiO<sub>2</sub>/Si structure by magnetron sputtering of a silicide target show that manganese silicide layers have a uniform fine-grained structure of a semiconductor nature, which is characterized by thermal sensitivity up to 20-30  $\mu$ V per degree.

Heated films Mn<sub>4</sub>Si<sub>7</sub> -146 nm coating has a uniform structure with fine grains, due to sufficient coating density. With an increase in temperature from room temperature to 800 K, a decrease in resistance is observed from 250 Ohms to 25 Ohms, which in turn indicates a change in the characteristics of this material. The Seebeck coefficient decreases upon heating from room temperature from 300 K to 620 K. Since Mn<sub>4</sub>Si<sub>7</sub> nanoclusters are semiconductor materials, it can be assumed that there will be energy barriers for charge carriers at the nanocluster–amorphous phase interface separating this phase.

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#### References

1 Markov, V.F. (2014). Materials of modern electronics. Ministry of Education and Science Ros. Federation. P. 272.

2 Bekpulatov, I.R., Imanova, G.T., Kamilov, T.S., Igamov, B.D. & Turapov, I.K. (2022). Formation of n — type CoSimonosilicide film which can be used in instrumentation. *International Journal of Modern Physics B.*, 2350164. DOI: 10.1142/S0217979223501643.

3 Umirzakov, B.E., Bekpulatov, I.R., Turapov, I.Kh. & Igamov, B.D. (2022). Effect of Deposition of Submonolayer Cs Coatings on the Density of Electronic States and Energy Band Parameters of CoSi2/Si(111). *Journal of Nano- and Electronic Physics, Vol. 14, 2, P. 02026.* DOI: 10.21272/jnep.14(2).02026.

4 Normurodov, M.T., Rysbaev, A.S., Bekpulatov, I.R., Normurodov, D.A. & Tursunmetova, Z.A. (2021). Formation and Electronic Structure of Barium-Monosilicide- and Barium-Disilicide Films. *Journal of Surface Investigation, Vol. 15*, P. S211-S215. DOI: 10.1134/S1027451022020318.

5 Isakhanov, Z.A., Umirzakov, Y.E., Ruzibaeva, M.K. & Donaev, S.B. (2015). Effect of the O2+-ion bombardment on the TiN composition and structure. *Technical Physics, Vol. 60, 2,* P. 313-315. DOI: 10.1134/S1063784215020097.

6 Umirzakov. B.E., Donaev, S.B. & Mustafaeva, N.M. (2019). Electronic and Optical Properties of GaAlAs/GaAs Thin Films. *Technical Physics, Vol. 64, 10*, P. 1506-1508. DOI: 10.1134/S1063784219100220.

7 Donaev, S.B., Umirzakov, B.E. & Mustafaeva, N.M. (2019). Emissivity of Laser-Activated Pd–Ba Alloy. *Technical Physics, Vol. 64, 10, P. 1541-1543.* DOI: 10.1134/S1063784219100074

8 Weissmuller, J. (1996). Synthesis and Processing of Nanocrystalline Powder / TMS.

9 Hicks, L.D. & Dresselhaus, M.S. (1993). Effect of quantum-well structures on thethermoelectric figure of merit. *Physical Review B.*, Vol. 47, P. 12727–12731.

10 Allon, I. (2008). Enhanced thermoelectric performance of rough silicon nanowires. NAT, Vol. 451, 7175, P.163–167.

11 Burkov, A.T., Novikov, S.V., & Schumann J. (2012). Nanocrystallization of AmorphousM-Si Thin Film Composites (M=Cr, Mn) and Their Thermoelectric Properties. *AIP Conference Proceedings, 9th European Conference on Thermoelectric, Vol. 1449,* P. 219-222.

12 Novikov, S.V., Burkov, A.T. & Schumann, J. (2013). Enhancement of thermoelectricproperties in nanocrystalline M–Si thin film composites (M = Cr, Mn). *Journal of Alloys and Compounds, Vol. 557,* P. 239–243.

13 Orekhov, A.S. & Klechkovskaya, V.V. (2017). Establishment of the relationship between the microstructure and thermoelectric properties of crystals of higher manganese silicide, doped with germanium. *Physics and technology of semiconductors, Vol. 51, 7, P.* 925-928.

14 Adachi K., Ito, K., Zhang, L.T. & Yamaguchi, M. (2003). Material Science. P. 3445–3450.

15 Maex, K. (2003). Properties of Metal Silicides. INSPEC, Stevenage.

16 Yoneyama, T. & Okada, A. (2013). Formation of polycrystalline BaSi2 films by radio-frequency magnetron sputtering for thin-film solar cell applications. *Thin Solid Films, Vol. 534*, P. 116-119. DOI: 10.1016/j.tsf.2013.02.003.

17 Dubov, V.L. (2008). Formation, structure, optical and photoelectric properties of textured BaSi2 films on Si(111) and heterostructures based on them. *Dissertation. Blagoveshchensk. Vol. 124*, P. 67-70.

18 Egerton, R.F. (2008). Electron energy-loss spectroscopy in the TEM. Reports on Progress in Physics. Vol. 72, 1, P. 016502.

19 Pani, M. & Palenzona, A. (2008). The phase diagram of the Ba-Si system. Journal of Alloys and Compounds, Vol. 454, 1-2, P. L1-L2.

20 Miyazaki, Y., Igarashi, D., Hayashi, K. & Kajitani, T. (2008). Physical Review B., Vol. 78, P. 214104.

21 Kajitani, T. (2010). Journal of Electronic Materials, Vol. 39, P. 1482-1487.

22 Migas, D.B. (2008). Physical Review B., Vol. 77, P. 075205.

23 Kamilov, T.S., Rysbaev, A.S., Klechkovskaya, V.V., Orekhov, A.S., Igamov, B.D. & Bekpulatov, I.R. (2019). The Influence of Structural Defects in Silicon on the Formation of Photosensitive Mn4Si7–Si(Mn)–Mn4Si7 and Mn4Si7–Si(Mn)–M Heterostructures. *Applied Solar Energy, Vol. 55*, P. 380-384.

#### Б.Д. Игамов, Г.Т. Иманова, А.И. Камардин, И.Р. Бекпулатов

# Нысанды қалыптастыру және магнетронды шашырату арқылы алынған Mn4Si7 жабындарын зерттеу

SiO<sub>2</sub>/Si құрылымында магнетронды шашырату арқылы алынған Mn4Si7 көлемдi үлгiлерi мен вакуумдық жабындарының морфологиясы, құрамы және электрлiк және оптикалық қасиеттерi зерттелдi. Қалыңдығы шамамен 150 нм болатын марганецтi силицидтi жабындар қасиеттерi бойынша массалық Mn4Si7-ге жақын, жартылай өткiзгiш сипаттағы бiркелкi ұсақ түйiршiктi құрылымы бар, ол бiр градусқа 20-30 мкВ-қа дейiнгi термиялық сезiмталдықпен сипатталады. Сонымен қатар мақалада авторлар магнетронды шашырату әдiсiмен шығарған жоғары марганецтi силицидтi қабыршақтардың электрофизикалық қасиеттерiн ұсынған. Қыздырылған пленкалар Mn4Si7-146 нм жабынның жеткiлiктi тығыздығына байланысты ұсақ түйiршiктерi бар бiркелкi құрылымға ие. Mn4Si7 нанокластерлерi жартылай өткiзгiш материалдар болғандықтан, осы фазаны бөлетiн нанокластер-аморфты фаза интерфейсiнде заряд тасымалдаушылар үшiн энергетикалық кедергiлер болады деп болжауға болады. Термиялық сезiмталдықтың 0 мкВ/К-ден 20 мкВ/К-ден 800 К-ге дейiн жоғарылауына нанокластерлердiң реттелгенiне байланысты нанокластер-аморфты фаза интерфейсiнде заряд тасымалдаушылар үшiн энергетикалық кедергiлер болады деп болжауға болады. Термиялық сезiмталдықтың 0 мкВ/К-ден 20 мкВ/К-ден 800 К-ге дейiн жоғарылауына нанокластерлердiң реттелгенiне байланысты нанокластер-аморфты фаза интерфейсiндегi заряд тасымалдаушыларға арналған энергетикалық кедергiлердiң жойылуымен түсiндiрiледi. Салқындату кезiнде 20 мкВ/К-ден 28 мкВ/К-ге дейiн өзгеруi аморфты фазадағы құрылымдық релаксацияның пайда болуымен түсiндiрiледi.

*Кілт сөздер:* Холл тұрақтысы, Мп4Si7, жұқа жабын, нанокластер, электрөткізгіштік, наноқұрылым, меншікті кедергі, көлемді концентрациясы.

## Б.Д. Игамов, Г.Т. Иманова, А.И. Камардин, И.Р. Бекпулатов

## Формирование мишеней и исследование покрытий Mn4Si7, полученных методом магнетронного распыления

Исследованы морфология, состав, электрические и оптические свойства объемных образцов и вакуумных покрытий Mn4Si7, полученных методом магнетронного напыления на структуру SiO<sub>2</sub>/Si. Показано, что покрытия из силицида марганца толщиной около 150 нм близки по свойствам к объемному Mn4Si7, имеют однородную мелкозернистую структуру полупроводниковой природы, которая характеризуется термочувствительностью до 20–30 мкВ на градус. Кроме того, в настоящей статье представлены электрофизические свойства пленок высокомарганцевого силицида, полученных авторами методом магнетронного распыления. Нагретые пленки Mn4Si7 — 146 нм. Покрытие имеет однородную структуру с мелкими зернами, что обусловлено достаточной плотностью покрытия. Поскольку нанокластеры Mn4Si7 являются полупроводниковыми материалами, можно предположить, что на границе раздела «нанокластер—аморфная фаза», разделяющей эту фазу, будут существовать энергетические барьеры для носителей заряда. Увеличение термочувствительности от 0 до 20 мкВ/К вплоть до 800 К объясняется исчезновением энергетических барьеров для носителей заряда на границе «нанокластер—аморфная фаза» за счет упорядочения нанокластеров. Изменение от 20 до 28 мкВ/К при охлаждении объясняется появлением структурной релаксации в аморфной фазе.

Ключевые слова: постоянная Холла, Mn4Si7, тонкое покрытие, нанокластер, электропроводность, наноструктура, удельное сопротивление, объемная концентрация.